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# On the Partition of the Strain-Rate into Elastic and Plastic Parts and Corresponding Constitutive Expressions

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By using Lee's intermediate configuration, partition of the strain-rate into elastic and plastic parts is analysed, associated with the Jaumann and convected rates of stress, as well as the corresponding constitutive structures.

## 1. Introduction

Let  $E$  be an objective, symmetric strain tensor and  $T$  its conjugate stress,  $T : dE$  being increment of work per unit volume in the reference state from where  $E$  is measured ( $:$  denotes the trace), [HR]. Let further  $\mathcal{L}^e$  be the corresponding instantaneous elastic moduli tensor, such that  $\mathcal{L}^e : dE$  is the stress increment that would result if the response on arbitrary strain variation  $dE$  were purely elastic. With  $\mathcal{M}^e = \mathcal{L}^e{}^{-1}$  as the instantaneous elastic compliance tensor,  $\mathcal{M}^e : dT$  is the strain increment that would result from purely elastic response corresponding to stress increment  $dT$ , and subtraction from the actual strain increment  $dE$  gives the plastic part  $d^p E = dE - \mathcal{M}^e : dT$ . This is the residual strain increment in an infinitesimal loading-unloading cycle of stress  $T$ , which can be shown to be governed by the plastic potential, i.e. codirectional with the outward normal to a locally smooth yield surface in stress  $T$  space. For example, by identifying  $E$  as the logarithmic strain, and with current state as a reference:  $T = \sigma$  (Cauchy stress),  $dE = D dt$  ( $D$  is the strain-rate and  $t$  a timelike parameter), and  $dT = \overset{\circ}{\tau} dt$ , where  $\overset{\circ}{\tau} = \overset{\circ}{\sigma} + (\text{tr } D)\sigma$  is the Jaumann rate of the Kirchhoff

stress  $\tau = (\det F)\sigma$  at current state as a reference (deformation gradient  $F$  being identity,  $F \equiv I$ ). Plastic contribution  $D^p$  to total strain-rate  $D$  is therefore

$$D^p = D - \mathcal{M}^e : \overset{\circ}{\tau}. \quad (1)$$

With different choice of conjugate variables and, therefore, different objective stress-rate, elastic compliance changes and what appears to be elastic and plastic strain-rates depends on the chosen stress-rate. Indeed, let  $E$  be the Lagrange strain, the conjugate stress is then symmetric Piola-Kirchhoff stress, and by taking the reference state coincident with the current state, we have:  $dE = D dt$  and  $dT = \overset{\circ}{\tau} dt$ , where  $\overset{\circ}{\tau} = \overset{\circ}{\tau} - D\sigma - \sigma D$  is convected derivative of  $\tau$  at  $F = I$ . Hence, in this case

$$\tilde{D}^p = D - \tilde{\mathcal{M}}^e : \overset{\circ}{\tau}, \quad (2)$$

To obtain, in a consistent manner, instantaneous elastic compliance ( $\mathcal{M}^e$  or  $\tilde{\mathcal{M}}^e$ ), the finite elasticity law

$$\tau = 2F^e \frac{\partial w(C^e)}{\partial C^e} F^{eT} \quad (3)$$

can conveniently be used.  $F^e$  is the elastic part of total deformation gradient, corresponding to elastic stressing from intermediate (stress free, relaxed) to current configuration, as given by [Lee] multiplicative decomposition

$$F = F^e F^p, \quad (4)$$

$F^p$  denoting the plastic part. In (3) plastic incompressibility is assumed,  $C^e$  is the right Cauchy-Green deformation tensor, while  $w$  is the strain energy per unit unstressed volume. For example, assuming elastic isotropy, which remains preserved during elasto-plastic deformation,  $w$  is an isotropic function of  $C^e$ . By using (4), the velocity gradient  $L = \dot{F}F^{-1}$  can be expressed as

$$L = \dot{F}^e F^{e-1} + F^e (\dot{F}^p F^{p-1}) F^{e-1}, \quad (5)$$

and by taking symmetric and antisymmetric parts, the velocity strain and spin tensors are:

$$D = (\dot{F}^e F^{e-1})_s + [F^e (\dot{F}^p F^{p-1}) F^{e-1}]_s, \quad (6)$$

$$W = (\dot{F}^e F^{e-1})_a + [F^e (\dot{F}^p F^{p-1}) F^{e-1}]_a. \quad (7)$$

Identification (separation) of elastic and plastic contributions on the right hand side of (6) has been the subject of many papers. We address this issue here from two perspectives: based on the Jaumann and convected derivatives used to define elastic strain rate.

## 2. Analysis based on the Jaumann derivative

As discussed in [Lub] any intermediate configuration can be used in the analysis. Often utilized choice is based on destressing without rotation ([Lee], [LuL]). A convenient choice is also obtained by destressing such that

$$[F^e (\dot{F}^p F^{p-1}) F^{e-1}]_a = 0, \quad (8)$$

as it turns out that in this case elastic and plastic parts of  $D$  are:

$$D^e = (\dot{F}^e F^{e-1})_s, \quad (9)$$

$$D^p = F^e (\dot{F}^p F^{p-1}) F^{e-1}. \quad (10)$$

Indeed, by differentiating (3)

$$\dot{\tau} = (\dot{F}^e F^{e-1})\sigma + \sigma(F^{e-T} \dot{F}^{eT}) + \frac{2}{\det F^e} F^e \left[ \frac{\partial^2 w(C^e)}{\partial C^e \otimes \partial C^e} : \dot{C}^e \right] F^{eT}, \quad (11)$$

we recognize (9), as equation (11) can then be written as

$$\overset{\circ}{\tau} = D^e \sigma + \sigma D^e + \frac{4}{\det F^e} F^e \left[ \frac{\partial^2 w(C^e)}{\partial C^e \otimes \partial C^e} : (F^{eT} D^e F^e) \right] F^{eT} \equiv \mathcal{L}^e : D^e. \quad (12)$$

This identifies elastic moduli tensor  $\mathcal{L}^e$ , or its inverse  $\mathcal{M}^e$ , appearing in (1) and corresponding to Jaumann derivative  $\overset{\circ}{\tau}$ . Note that for any other choice of intermediate configuration (not defined by (8)), symmetric part of  $\dot{F}^e F^{e-1}$  is not all, but just a portion of the elastic strain-rate  $D^e$ . Elastic moduli  $\mathcal{L}^e$  and elastic strain-rate  $D^e$  are, of course, independent of the selected intermediate configuration. These issues are in detail discussed in [Lub].

### 3. Analysis based on the convected derivative

#### 3.1 Geometric preliminaries

Introduce the convected coordinates  $\theta^i$ , superscripted indices denote contravariant components and subscripted the covariant components [Hut]. Denote base vectors in the undeformed state by  $\mathbf{g}_i$  and their reciprocals by  $\mathbf{g}^i$ . The base vectors in the deformed (current and intermediate) states are denoted by  $\bar{\mathbf{g}}_i$  and  $\hat{\mathbf{g}}_i$ , respectively, their reciprocals being  $\bar{\mathbf{g}}^i$  and  $\hat{\mathbf{g}}^i$ . The metric tensors of the three configurations are therefore:  $g_{ij} = \mathbf{g}_i \cdot \mathbf{g}_j$ ,  $\bar{g}_{ij} = \bar{\mathbf{g}}_i \cdot \bar{\mathbf{g}}_j$  and  $\hat{g}_{ij} = \hat{\mathbf{g}}_i \cdot \hat{\mathbf{g}}_j$ , and let  $g^{ij}$ ,  $\bar{g}^{ij}$  and  $\hat{g}^{ij}$  be their respective inverses. From the definition of the intermediate configuration:  $\bar{\mathbf{g}}_i = F \mathbf{g}_i = F^e \hat{\mathbf{g}}_i$  and  $\hat{\mathbf{g}}_i = F^p \mathbf{g}_i$ .

The Lagrangian ( $\eta$ ) strain tensor is defined by:

$$\eta = \eta_{ij} \mathbf{g}^i \otimes \mathbf{g}^j, \quad \eta_{ij} = \mathbf{g}_i \cdot \eta \cdot \mathbf{g}_j = \frac{1}{2}(\bar{g}_{ij} - g_{ij}) \quad (13)$$

which also can be expressed via deformation tensor  $C = F^T F$  as  $\eta = \frac{1}{2}(C - I)$ . Now, if  $dS$ ,  $ds$  and  $d\hat{s}$  are the lengths of an infinitesimal material element in the initial, current and intermediate state, it follows:

$$ds^2 - dS^2 = (\bar{g}_{ij} - g_{ij})d\theta^i d\theta^j = (\bar{g}_{ij} - \hat{g}_{ij})d\theta^i d\theta^j + (\hat{g}_{ij} - g_{ij})d\theta^i d\theta^j. \quad (14)$$

Consequently, we define elastic and plastic parts of Lagrange strain by:

$$\begin{aligned} \eta^e &= \eta_{ij}^e \mathbf{g}^i \otimes \mathbf{g}^j, & \eta_{ij}^e &= \frac{1}{2}(\bar{g}_{ij} - \hat{g}_{ij}) \\ \eta^p &= \eta_{ij}^p \mathbf{g}^i \otimes \mathbf{g}^j, & \eta_{ij}^p &= \frac{1}{2}(\hat{g}_{ij} - g_{ij}) \end{aligned} \quad (15)$$

and from (13) and (15),  $\eta = \eta^e + \eta^p$ . On the other hand

$$ds^2 - d\hat{s}^2 = d\mathbf{x} \cdot d\mathbf{x} - d\hat{\mathbf{x}} \cdot d\hat{\mathbf{x}} = d\hat{\mathbf{x}} \cdot (C^e - I) \cdot d\hat{\mathbf{x}} = \hat{\mathbf{g}}_i \cdot (C^e - I) \cdot \hat{\mathbf{g}}_j d\theta^i d\theta^j, \quad (16)$$

hence, in view of  $ds^2 - d\hat{s}^2 = 2\eta_{ij}^e d\theta^i d\theta^j$ ,

$$\eta_{ij}^e = \hat{\mathbf{g}}_i \cdot \hat{\boldsymbol{\eta}}^e \cdot \hat{\mathbf{g}}_j, \quad \hat{\boldsymbol{\eta}}^e = \frac{1}{2}(C^e - I). \quad (17)$$

Expressing  $\hat{\boldsymbol{\eta}}^e$  via its components on  $\hat{\mathbf{g}}^i$  reciprocal basis

$$\hat{\boldsymbol{\eta}}^e = \hat{\eta}_{ij}^e \hat{\mathbf{g}}^i \otimes \hat{\mathbf{g}}^j = \frac{1}{2}(C_{ij}^e - \hat{g}_{ij})\hat{\mathbf{g}}^i \otimes \hat{\mathbf{g}}^j \quad (18)$$

and comparison with (17) gives

$$\eta_{ij}^e \equiv \hat{\eta}_{ij}^e = \frac{1}{2}(C_{ij}^e - \hat{g}_{ij}). \quad (19)$$

Introducing further the strain-rate  $D$ , it follows that

$$\bar{\mathbf{g}}_i \cdot D \cdot \bar{\mathbf{g}}_j \equiv \mathbf{g}_i \cdot \dot{\boldsymbol{\eta}} \cdot \mathbf{g}_j = \dot{\eta}_{ij} \quad (20)$$

and, therefore, by writing  $D = D_{ij} \bar{\mathbf{g}}^i \otimes \bar{\mathbf{g}}^j$ , we have  $D_{ij} \equiv \dot{\eta}_{ij}$ . Substitution of  $D = \bar{D}^e + \bar{D}^p$  into (20) then gives

$$\bar{\mathbf{g}}_i \cdot (\bar{D}^e + \bar{D}^p) \cdot \bar{\mathbf{g}}_j = \dot{\eta}_{ij}^e + \dot{\eta}_{ij}^p. \quad (21)$$

In the following subsection we shall show that elastic and plastic strain-rates, corresponding to convected stress rate  $\overset{\square}{\tau}$ , turn out to be:

$$\bar{D}^e = \dot{\eta}_{ij}^e \bar{\mathbf{g}}^i \otimes \bar{\mathbf{g}}^j \quad (22)$$

$$\bar{D}^p = \dot{\eta}_{ij}^p \bar{\mathbf{g}}^i \otimes \bar{\mathbf{g}}^j. \quad (23)$$

### 3.2 Constitutive expression for elastic strain-rate

The elastic part of deformation gradient  $F^e$  can be expressed in dyadic form with respect to intermediate basis  $\hat{\mathbf{g}}_i$  as

$$F^e = (F^e)^{ij} \hat{\mathbf{g}}_i \otimes \hat{\mathbf{g}}_j, \quad (24)$$

and substitution into (3) gives, after easy manouvers,

$$J \sigma^{ij} \bar{\mathbf{g}}_i \otimes \bar{\mathbf{g}}_j = 2 \frac{\partial w}{\partial C_{ij}^e} \bar{\mathbf{g}}_i \otimes \bar{\mathbf{g}}_j, \quad \text{i.e.} \quad J \sigma^{ij} = 2 \frac{\partial w}{\partial C_{ij}^e} \quad (25)$$

with  $J$  standing for  $\det F$ , while  $\sigma^{ij}$  are the contravariant components of the Cauchy stress on the current basis  $\bar{\mathbf{g}}_i$ . Utilizing Eq. (19), above can be rewritten in the form

$$J \sigma^{ij} = \frac{\partial w}{\partial \eta_{ij}^e} \quad (26)$$

and differentiation gives

$$(\text{tr } D) \sigma^{ij} + \dot{\sigma}^{ij} = \frac{1}{J} \frac{\partial w^2}{\partial \eta_{ij}^e \partial \eta_{kl}^e} \dot{\eta}_{kl}^e. \quad (27)$$

differentiating the Kirchhoff stress

Next, by

$$\tau = \tau^{ij} \bar{\mathbf{g}}_i \otimes \bar{\mathbf{g}}_j \quad (28)$$

and by using  $\dot{\bar{\mathbf{g}}}_i = L \cdot \bar{\mathbf{g}}_i$ ,  $L = \dot{F}F^{-1} = \dot{\bar{\mathbf{g}}}_k \otimes \bar{\mathbf{g}}^k$  being the velocity gradient, we obtain

$$\dot{\tau}^{ij} \bar{\mathbf{g}}_i \otimes \bar{\mathbf{g}}_j = \dot{\tau} - L\tau - \tau L^T, \quad (29)$$

which is the convected derivative of the Kirchhoff stress, associated with the choice  $F = I$ . Hence, we recognize (22), since expression (27) then becomes

$$\overset{\circ}{\tau} = \tilde{\mathcal{L}}^e : \tilde{D}^e, \quad (\tilde{\mathcal{L}}^e)^{ijkl} = \frac{1}{J} \frac{\partial^2 w}{\partial \eta_{ij}^e \partial \eta_{kl}^e} \quad (30)$$

or by inversion

$$\tilde{D}^e = \tilde{\mathcal{M}}^e : \overset{\circ}{\tau}, \quad \tilde{\mathcal{M}}^e = \tilde{\mathcal{L}}^e{}^{-1}. \quad (31)$$

This establishes the elastic compliance tensor appearing in (2), corresponding to convected derivative  $\overset{\circ}{\tau}$ . If elastic component of strain is small,  $\mathcal{L}^e \approx \tilde{\mathcal{L}}^e$  and  $\mathcal{M}^e \approx \tilde{\mathcal{M}}^e$ .

#### 4. Discussion

Having established constitutive expression for the elastic strain-rate, one needs to derive expression for the associated plastic strain-rate. When the Jaumann derivative  $\overset{\circ}{\tau}$  is utilized, development goes in a straightforward manner. For example, in the case of isotropic hardening and Mises yield condition, one obtains

$$D^p = \dot{\lambda} \sigma', \quad \dot{\lambda} = \frac{1}{h} \sigma' : \overset{\circ}{\tau} \quad (32)$$

( $h$  is a hardening parameter), which expresses the normality to Mises yield surface and coaxiality of  $D^p$  with the deviatoric part of stress  $\sigma'$ . Rewriting (32)



in the form  $D^p = \mathcal{M}^p : \overset{\circ}{\dot{\tau}}$  and combining with  $D^e = \mathcal{M}^e : \overset{\circ}{\dot{\tau}}$  one arrives at the well known elasto-plastic constitutive structure, such as for example given in [MR]. However, when the convected derivative  $\overset{\circ}{\dot{\tau}}$  is utilized in the partition of the strain-rate into corresponding elastic and plastic parts, an explicit development of the constitutive expression for strain-rate  $\overset{\circ}{\dot{D}}^p$  becomes more difficult. For example,  $\overset{\circ}{\dot{D}}^p$  is not normal to Mises yield surface, so that modification of the associated flow rule or the yield condition is needed in order to obtain a physically plausible constitutive law. Of course, the transition from the final elasto-plastic constitutive structure with the Jaumann derivative  $\overset{\circ}{\dot{\tau}} = \mathcal{L} : D$ , to one with the convected derivative is direct via  $\overset{\circ}{\dot{\tau}} = \overset{\circ}{\dot{\tau}} + D\sigma + \sigma D$ .